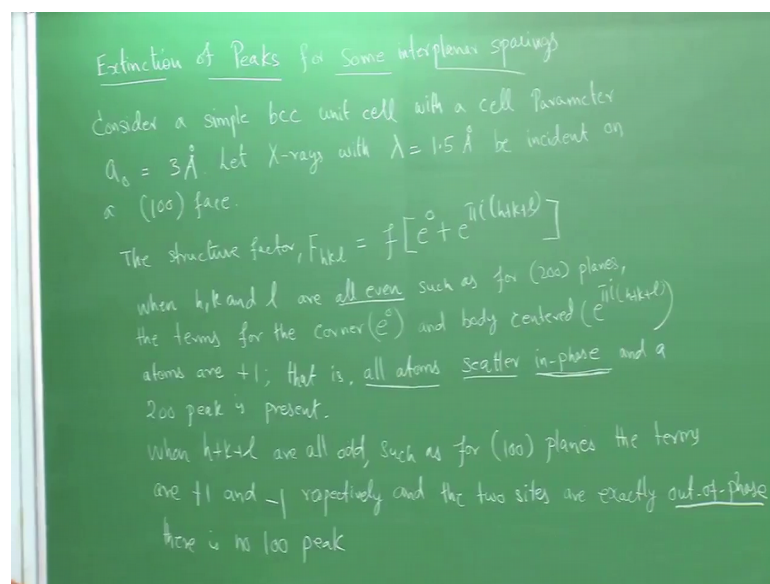


X-Ray Crystallography
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Lecture - 18
Tutorial 04

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Now we will look at the extinction of a peaks for some inter planar spacings; in this case we will take BCC unit cell I consider a simple BCC unit cell with a cell parameter a naught equal to 3 angstrom let X-rays with wavelength lambda which is equal to 1.5 angstrom be incident on the 1 0 0 phase. Now, we will calculate the structure factor. So, this is the structure factor equation because of the 2 atoms were unit cell. So, when h k l are all even fetches for 2 0 0 planes.

For example, the terms for the corner atoms that is e power 0 and the body centred atom e power pi i into h plus k plus l atoms are plus one; that means, what does that mean; that means, all atoms scatter in phase and 2 0 0 peak is present. So, according to this rule all atoms scatters in phase. So, these things you have remember and 2 0 0 peak is also present.

However, when h , k and l are all even such as $1\ 0\ 0$ plane the terms are plus 1 and minus 1 respectively and the 2 sides exactly out of phase when there is no $1\ 0\ 0$. In fact, what I have written is what I have shown in the slide in the previous slide where I showed the reciprocal lattice where you could see only these kinds of planes.

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From Bragg's law for scattering from (100) planes
 $\lambda = 2d_{100} \sin \theta_{100}$ from which $\sin \theta = \frac{1.5}{2 \times 3} = 0.25$
 and $\theta = 14.48^\circ$
 The difference in the path length between rays 1 and 2
 is obtained from $\sin \theta_{100} = \frac{AC}{d_{100}} = \frac{BC}{d_{100}} = 1.5 \text{ \AA}$
 which is one X-ray wavelength
 (partially visible text on the left: "nd q", "e term", "lly out-of-phase")

These kind of plane because of this extinction rules now we can prove this concept geometrically we first solve Bragg's law for scattering from Bragg's law we write for scattering $1\ 1$ planes λ is equal to $2d_{1\ 0\ 0}$.

Let us understand what we have done. So, from the Bragg law for scattering from $1\ 0\ 0$ planes you can simply write λ is equal to $2d_{1\ 0\ 0}$ plane $\sin \theta_{1\ 0\ 0}$ from which we can find out the $\sin \theta$ value $\sin \theta$ is equal to 1.5 divided by 2 into 3 which is equal to 0.25 correspond to the $\theta = 14.48^\circ$; the difference in the path length between rays 1 and 2 the of the rays shown in the side you can see that we can obtain it from $\sin \theta_{1\ 0\ 0}$ equal to a/c divided by $d_{1\ 0\ 0}$ which is equal to BC divided by $d_{1\ 0\ 0}$ which is equal to 1.5 angstrom which is one X-ray wavelength. So, that is what we have stated with.

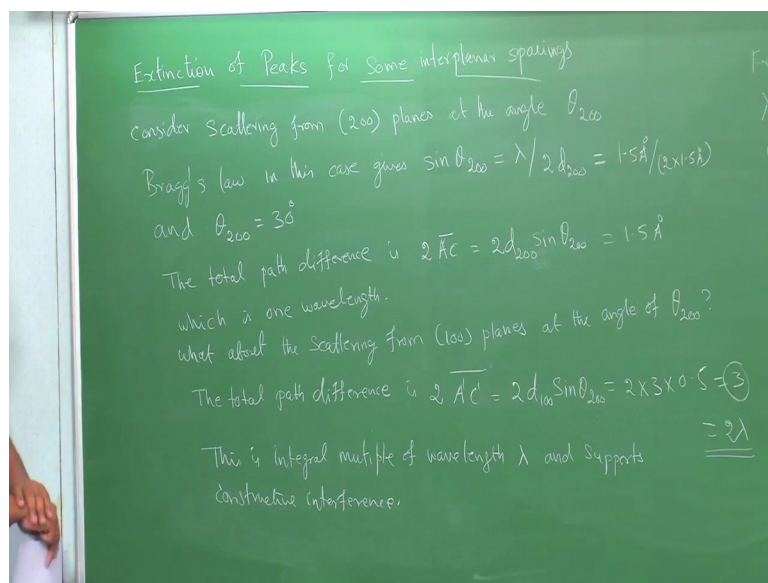
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$\lambda = 2d_{100} \sin \theta_{100}$ from which $\sin \theta = 1.5 / (2 \times 3) = 0.25$
 and $\theta = 14.48^\circ$
 The difference in the path length between rays 1 and 2
 is obtained from $\sin \theta_{100} = \overline{AC} / d_{100} = \overline{BC} / d_{100} = 1.5 \text{ \AA}$
 which is one X-ray wavelength
 What about the X-rays scattering from (200) planes at the
 angle θ_{100} ?
 The path length for (200) planes is $2 \overline{A'C'} = 2 d_{200} \sin \theta_{100}$
 which is half a wavelength ($\lambda/2$) and the $= 2 \times 1.5 \text{ \AA} \times 0.25$
 Waves from these planes completely cancel with $= 0.75 \text{ \AA}$
 those from (100) planes.

So, it is 1 X-ray wavelength now the question is what is what about the X-rays scattering from 2 0 0 planes at the angle theta 1 0 0. So, we can find out in similar way the path length let me rewrite. So, the path length for 2 0 0 planes is 2 times A dash C dash which is equal to 2 d sin theta 1 0 0, we can substitute this you will get 0.75 as path length which is half of the;

So, what the consequence if you have the path length from 2 0 0 planes is exactly half the wavelength. As we have discussed previously it will completely cancel out the intensity we can write that which will. So, when you have the path length for 2 0 0 planes is as half a wavelength then the wave from these planes completely cancel with those from 1 0 0 planes that is why you do not see the peaks 1 0 0 3 0 0.

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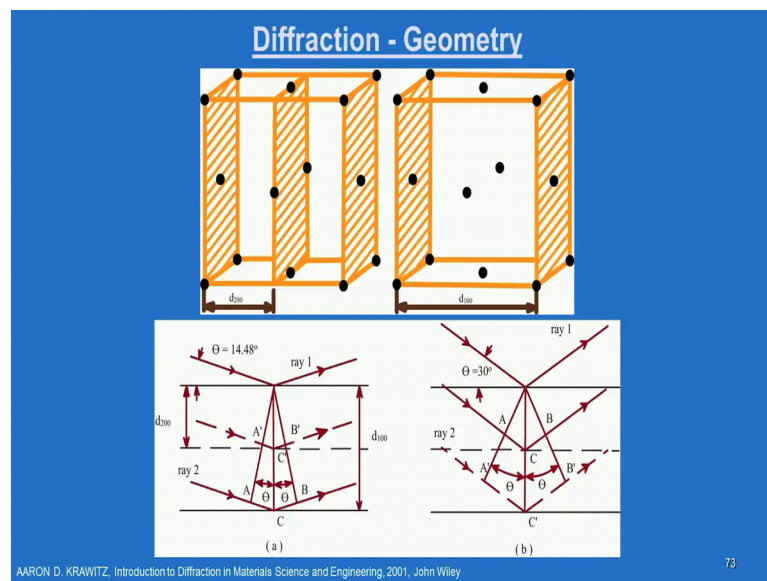
And. So, on in that reciprocal lattice now let us consider the scattering from. So, let us consider these scattering from 2 0 0 planes at the angle theta 2 0 0 if you look at the slide the figure b shows this schematic excuse me. And then we will proceed with the calculations and the Bragg law for this case gives sin theta 2 0 0 is equal to lambda by 2 d 2 0 0 which is nothing but 1.5 angstrom divided by 2 times 1.5 angstrom which is worked out to theta 2 0 0 is 30 degree.

Now, at this angle the total path difference the total path difference is 2 times a c which is 2 d 0 0 sin theta 2 0 0 which is nothing but 1.5 angstrom which is nothing but one wavelength. Now similar question what we have asked there what about scattering from. So, similar question here what about the scattering from 1 0 0 planes at the angle of theta 2 0 0. Now then we can work it out this also the total path difference. So, the total path difference in the case of scattering from 1 0 0 plane at the angle of theta 2 0 0 is 2 times a prime c prime which is equal to 2 d 1 0 0 sin theta 2 0 0 which is equal to 2 into 3 into point five which is nothing but 3 3 is 2 lambda this value is nothing but 2 lambda this is which is nothing but an integral multiple of a wavelength length lambda and supports the constructive inter prints process.

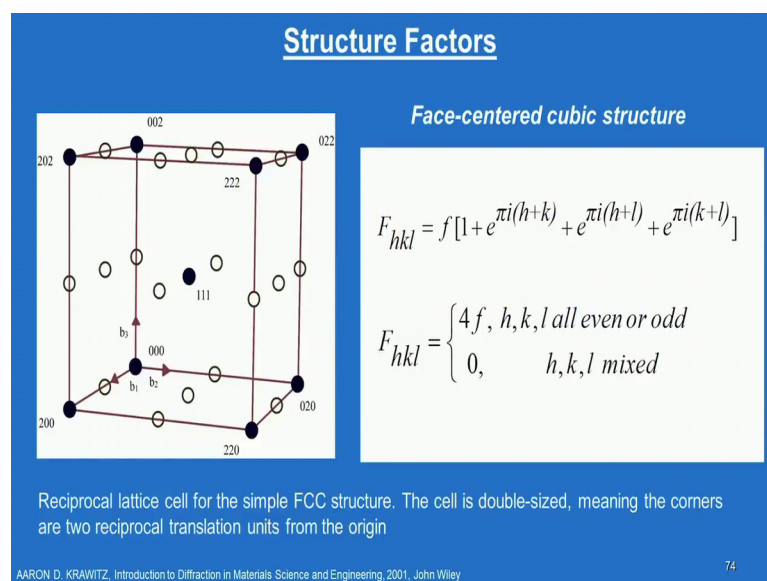
So, from this example we have seen how this 2 0 0 type of planes exist and what is the

meaning of extinction of certain peaks in for some inter planar spacing in crystal in cubic crystal like a BCC. So, these 2 examples give you some kind of an appreciation for this concept and now we will move on to the next example what I would like to show come back to this slides if you look at this slide this is structure factor calculation for a phase centred cubic structure.

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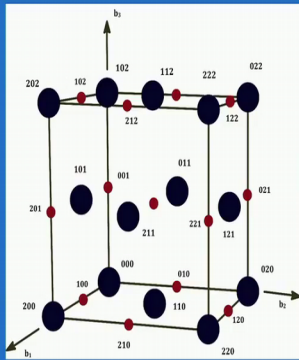


And if you write the expression you have atoms per unit cell. So, you get the 4 terms here. So, $1 + e^{i\pi h} + e^{i\pi k} + e^{i\pi l}$ to the power $i\pi(h+k+l)$. So, the rule is f_{hkl} is equal to $4f$ where h, k, l all even or odd and the intensity will become our structure factor will become 0 when h, k, l are mixed.

So, now you see that this is again a reciprocal lattice cell. So, do not confuse this with your real cell this is reciprocal unit cell. That means, you will see the intensity spot actually in a double cell size and wherever you have the allowed reflection then only you will see the dark circles. So, here you can see that all h, k, l all even or all odd will have the allowed reflections. So, other positions are forbidden. So, what you can now appreciate is the reciprocal unit cell or an FCC lattice actually a BCC lattice it does the same thing what we have seen before it is the other way around for a real BCC lattice the reciprocal lattice is an FCC unit cell. So, that is clearly appreciated here.

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Structure Factors



Cesium chloride (CsCl) structure

$$F_{hkl} = f_C + f_{Cs} e^{i\pi(h+k+l)}$$

$$F_{hkl} = \begin{cases} f_C + f_{Cs}, & h+k+l = 2n \\ f_C - f_{Cs}, & h+k+l = 2n+1 \end{cases}$$

The reciprocal lattice structure cell for the CsCl structure. The peaks for which $h+k+l$ is even are sum peaks and are bigger than those for which $h+k+l$ is odd

AARON D. KRAWITZ, Introduction to Diffraction in Materials Science and Engineering, 2001, John Wiley
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Now we will see some interesting examples a caesium chloride structure where you have a caesium chloride type of structure you can see that it is kind of body centred unit cell has got 2 atoms that is chlorine and caesium atoms per unit cell which is one corner another is in the body centred position, but you can write the expression in terms of the structure factor equation like this and the selection rule is you see f_{hkl} is equal to f_C

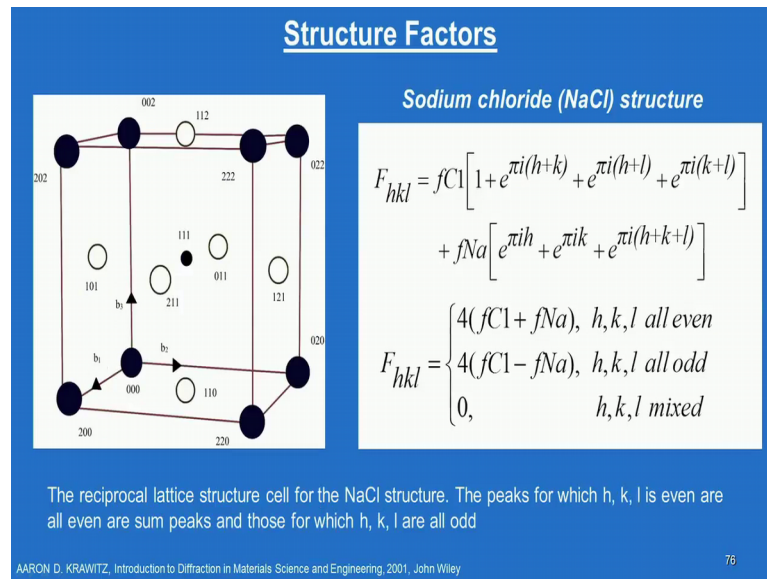
plus f_C where $h + k + l$ is equal to $2n$ and if it is f_C minus f_C where $h + k + l$ is equal to $2n + 1$.

See what you have to appreciate in this particular example is it is though it is BCC crystal it is the since it is chromium caesium are 2 different atoms it is to be considered as 2 BCC crystals are interpenetrating with each other it is an ordered system. And you should also appreciate that since these 2 atoms are very I mean different chlorine and caesium their scattering factor will be very different individual as an individual atom.

So, you will see instead of some reflections instead of a sum of the peak intensities you will see the difference in that scattering power that is why you see in some terms it is plus and some term it is minus here and the corresponding effect you will see in the bigger circle here. That means, higher intensity that is sum of intensity, and then you will have sum of these it is the difference peak this a different peak this is a sum peak that is what it is written the peaks for which $h + k + l$ is even or sum peaks. That means, when $h + k + l$ is equal to $2n$ means a sum peaks and are bigger than the those with $h + k + l$ is odd something like the red spot here which are all difference peak it is not the sum peak it is called a difference peak here.

And this is again a reciprocal lattice cell and you will be able to appreciate this when you look at the electron diffraction pattern much more clearly you will see these kind of an evidence actual practical evidence for the difference in the intensities. And this structure factor calculation nicely shows that the difference in the intensities, because of the factors like this.

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Now, we will move on to another important actors to system called a sodium chloride structure and this is similar to caesium chloride you have a 2 interpenetrating FCC crystal units. And you can see that the 2 positions are mentioned here one in I mean you can this is the final expression what we have written and you can see that f_{hkl} is equal to 4 times f_{Cl} plus f_{Na} where h, k, l are all even that is a sum peak you will see and there is a difference peak when h, k, l are all odd and if they are mixed there is the structure factor is 0.

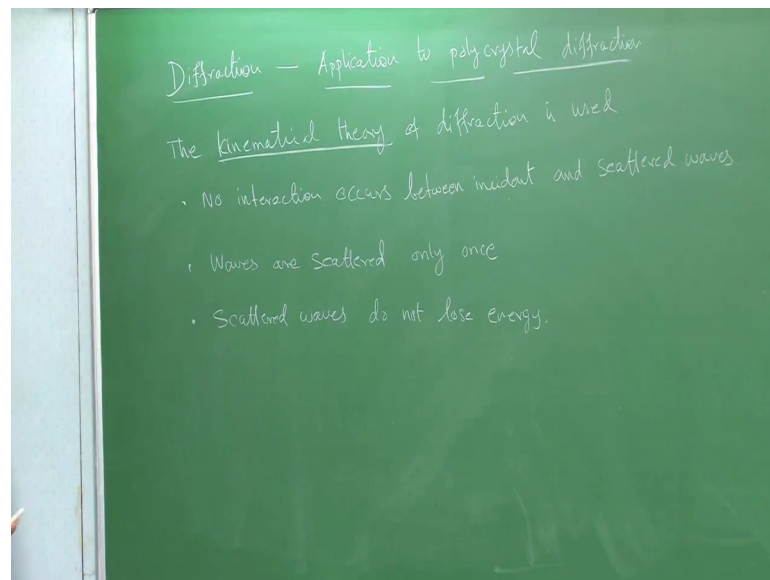
So, you can see this you can look at this selection rule and then corresponding the reciprocal lattice structure you can see that all some peak will have higher intensity that is larger circle dark circle and then there is a difference peak like 1 1 1 you will have a lesser intensity. And then you have a 0 intensity in h, k, l are all mixed like this. So, it is just to give an idea how the structure factor nicely gives you the idea of the intensity summing up or it is becoming 0 or it becoming at the difference in the scattering power of the atoms.

So, these illustrations clearly show they simply demonstrate the significance of the structure factor in appreciating the X-ray diffraction intensity. So, having said that we are only interested in appreciating the or understanding the intensity from the a single crystal

or it could be a poly crystal or it could be an amorphous material but we have to now look at the intensity expressions what are all the term it contains. So, before we really get into that we have to appreciate what are all the factors which influence the intensity that we have to look at it.

So, before I get into that discussion let me tell you the we have when you when you look at the X-ray diffraction spectrum in every peak there are 2 things you have I max and you have the breadth of the peak or the integrated intensity there are 2 these 3 things are very important I max is always not important, but it is the integrated intensity which is important which is the area under the whole peak and this is influenced by so many factors and this we have to understand one-by-one and then how the broadening of this intensity X-ray intensity peak is related to what we are looking at in the material. That is also an another important factor we will look at one-by-one. So, I will go back to the black board again.

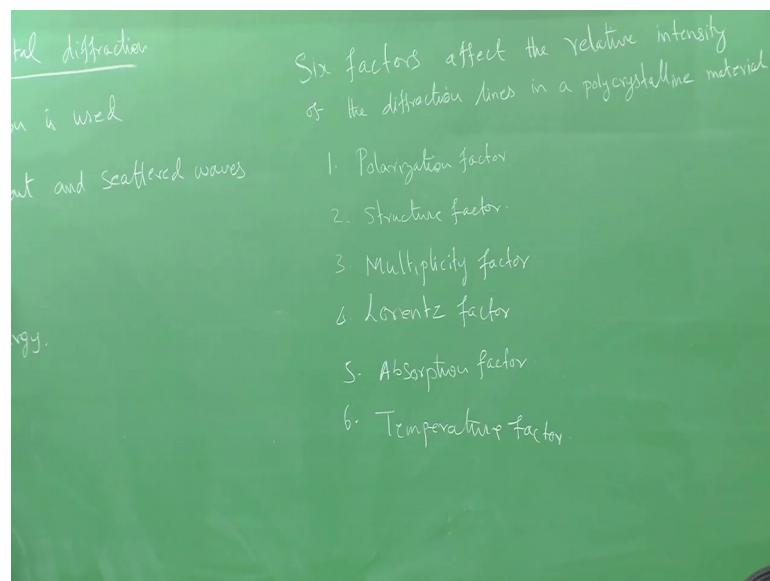
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So, before we talk about the intensity of the X-ray peak we should know what is the theory behind it when it when it being calculations are carried out and so, you see that the kinematical theory is used for calculating the intensity of X-ray and it has got some assumption which is very important. So, I will write down 1 1 3 important assumptions.

So, the kinematic theory of diffraction assumes these 3 points that is no interaction occurs between incident and the scattered waves. Secondly, waves are scattered only once. And thirdly, scattered waves loose energy. So, based on these assumptions the intensity of the X-ray is been calculated and then now I will just list out some of the factors which will influence the X-ray intensity.

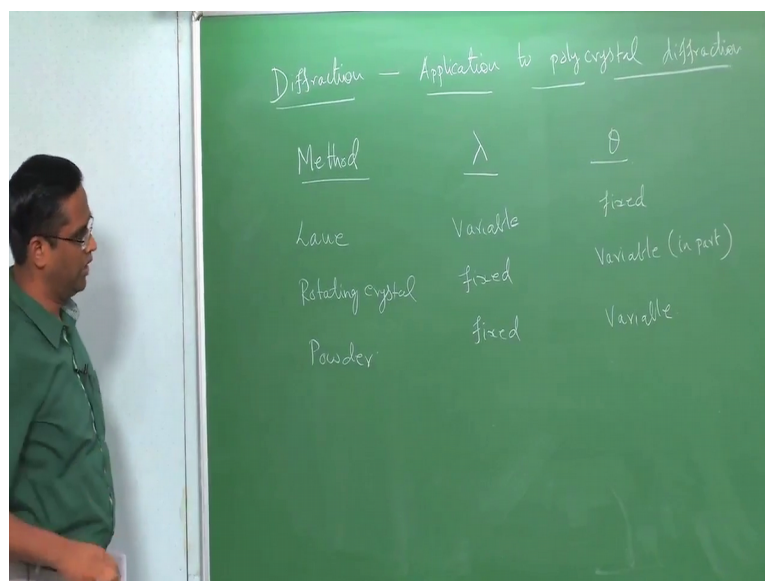
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There are 6 factors. So, there are 6 factors that will affect the relative intensity of the diffraction lines in a poly crystal material one is a polarization factor which we have already seen it you know you have some idea about what is polarization factor.

And then structure factor which we have now seen. So, at least these 2 are familiar the third one is multiplicity factor and fourth one is Lorentz factor fifth one is absorption factor and sixth one is temperature factor. So, all these factors have to be taken into account when you write a complete intensity expression for a diffracted beam. And before we get in to all these you have to remember that of course, this intensity also what you obtained is depending upon the method by which you perform these experiments there are primarily 3 methods that also I will write before we get in to the intensity discussion.

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Diffraction — Application to polycrystal diffraction

<u>Method</u>	<u>λ</u>	<u>θ</u>
Lave	Variable	fixed
Rotating crystal	fixed	Variable (in part)
Powder	fixed	Variable

So, there are 3 methods one is Lave method another is rotating crystal method and third is a powder method. I will not get into the theory of all these things. I will take you to the laboratory and then actually show what are all the typical methods we follow in the X-ray diffraction laboratory, when we actually perform the experiment just for the sake of completion. You should know the basic method of conducting this diffraction experiments when in a Lave condition or Lave method where the λ is variable.

And your θ is fixed in a rotating crystal method λ is fixed and your θ is variable partly and in a powder method you have a fixed λ . And then variable θ . These are all just about the methods by which you measure the X-ray diffraction intensity for polycrystalline material. So, we will look at the importance of the variable λ and fixed θ or fixed λ and variable θ when we do the laboratory demonstration.

So, now we will come back to the intensity of these X-rays. You have to have some idea about what all these factors mean. We will not get into the very detailed of the calculation of this or derivation of this which is out of scope of this course, but you should have some idea what is this factor because all these factors are going to come in the final intensity equation. And then we will briefly go through what is this multiplicity factor and Lorentz factor and absorption factor and temperature factor in the next class.

Thank you.